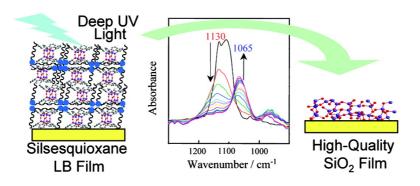


### Communication

## Photoinduced High-Quality Ultrathin SiO Film from Hybrid Nanosheet at Room Temperature

Yeji Kim, Feng Zhao, Masaya Mitsuishi, Akira Watanabe, and Tokuji Miyashita

*J. Am. Chem. Soc.*, **2008**, 130 (36), 11848-11849 • DOI: 10.1021/ja803852w • Publication Date (Web): 14 August 2008 Downloaded from http://pubs.acs.org on February 8, 2009



## **More About This Article**

Additional resources and features associated with this article are available within the HTML version:

- Supporting Information
- Access to high resolution figures
- Links to articles and content related to this article
- Copyright permission to reproduce figures and/or text from this article

View the Full Text HTML





#### Published on Web 08/14/2008

# Photoinduced High-Quality Ultrathin SiO<sub>2</sub> Film from Hybrid Nanosheet at Room Temperature

Yeji Kim, Feng Zhao, Masaya Mitsuishi,\* Akira Watanabe, and Tokuji Miyashita\*

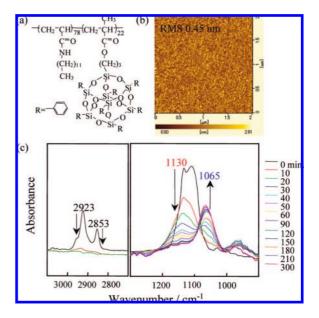
Institute of Multidisciplinary Research for Advanced Materials (IMRAM), Tohoku University, 2-1-1 Katahira, Aoba-ku, Sendai 980-8577, Japan

Received May 22, 2008; E-mail: masaya@tagen.tohoku.ac.jp; miya@tagen.tohoku.ac.jp

For developing industrial applications in the field of optoelectronics, development of a high-quality thin silicon dioxide (SiO<sub>2</sub>) film has been increasingly important<sup>1</sup> despite the increasing popularity of alternative materials.<sup>2</sup> Diverse silicon oxidation methods and mechanisms have been investigated extensively in recent decades.<sup>3</sup> Thermally oxidized SiO<sub>2</sub> films are now commercially available and are generally grown or annealed at high temperatures (about 1000 °C) to yield high quality. However, these thermal methods can severely damage the substrate and engender defect formation. Consequently, a few organic polymers have become attractive as alternatives to permit the use of a plastic substrate.<sup>4</sup> Nevertheless, the thin organic dielectric films must not only have excellent electrical properties, but also processability, low roughness, and low concentrations of impurities. Moreover, it is difficult to satisfy the demands of commercial applications such as high performance, high breakdown voltage, and long-term stability. Therefore, for developing flexible devices with plastic substrates, low-temperature preparation of SiO2 is urgently required.5

Results of prior studies have shown that organic—inorganic hybrid materials satisfy requirements such as ease of processability and compatibility. An ultrathin hybrid nanosheet containing poly-octahedral silsesquioxane (POSS) has been designed to prepare densely packed ultrathin POSS films using the Langmuir—Blodgett (LB)<sup>6</sup> technique. The LB technique enables POSS to have a multilayered structure with nanoscale precision. This nanoarchitecture is a good candidate for fulfilling further requirements for increasing device miniaturization. Herein, we demonstrate a flexible approach to building up high-quality ultrathin SiO<sub>2</sub> films using a deep UV lamp. An organic—inorganic hybrid nanosheet prepared using the LB technique<sup>7</sup> was used as a precursor for SiO<sub>2</sub> film preparation. Extremely smooth and robust SiO<sub>2</sub> film was produced at room temperature through photoirradiation.

The precursor hybrid material, poly(N-dodecylacrylamide-co-3methacryloxypropyl-T8-heptaphenyl POSS), p(DDA/SQ) was synthesized using free radical polymerization (Figure 1a). Details related to the synthesis of p(DDA/SQ) are available in a previous report.8 The p(DDA/SQ) nanosheet was deposited onto a substrate by vertical dipping with surface pressure at 30 mN/m using the LB technique. The absorbance of the nanosheet at 194 nm increases linearly with the increased number of deposited nanosheets (Supporting Information, Figure S1), underscoring the remarkable capability of a hybrid nanosheet to produce a multilayered structure with nanoscale precision. The surface morphology of the hybrid nanosheet showed quite a smooth surface and uniform distribution of POSS moieties in the p(DDA/SQ) nanosheets. The precursor p(DDA/SQ) nanosheets were irradiated to prepare a SiO<sub>2</sub> film. As illustrated by the AFM image depicted in Figure 1b, the photoirradiated film was flat (rms 0.45 nm in  $2 \times 2 \mu m^2$ ). The resulting film thickness, as measured using photoirradiation, was one-fifth



**Figure 1.** (a) Chemical structure of p(DDA/SQ); (b) tapping-mode AFM image of photooxidized SiO<sub>2</sub> film; (c) FTIR spectra of a p(DDA/SQ) hybrid nanosheet upon photoirradiation.

less than that of the pristine film, irrespective of the number of hybrid nanosheets. This predictable decrease in thickness is explainable by selective removal of organic moieties from the photoirradiated nanosheet. The photooxidized SiO2 film density was also determined using a quartz crystal microbalance (QCM): 2.24  $g/cm^3$ , which is comparable to that of a thermally oxidized SiO<sub>2</sub> film. Furthermore, the mechanical properties of hybrid nanosheets before and after photoirradiation were investigated using nanoindentation.9 The hardness and shear modulus of the films changed considerably, from 0.19 and 2.62 GPa to 1.74 and 32.22 GPa, respectively. These dramatic changes in the density and mechanical properties are probably attributable to the change in chemical structure by photochemical reaction as well as the dissociation of organic moieties in hybrid nanosheets. The salient implication is that a Si-O-Si network structure based on SiO<sub>2</sub> can be created using photooxidation.

Figure 1c depicts FTIR spectra of p(DDA/SQ) nanosheets during photoirradiation at room temperature. The bands at 2853 and 2923 cm<sup>-1</sup>, which are ascribed to symmetric and asymmetric vibrations of the C–H groups, decreased and subsequently disappeared with photoirradiation. Complete removal of DDA from the nanosheet by photoirradiation was also confirmed from the UV–vis spectra (Figure S3). The sharp absorption peak at 1130 cm<sup>-1</sup> based on Si–O–Si cage structure decreased with photoirradiation. Corresponding to the decrease of 1130 cm<sup>-1</sup> absorption peak, increased absorbance at 1065 cm<sup>-1</sup> was observed. The peak position is

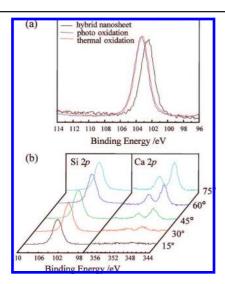


Figure 2. (a) XPS spectra of Si 2p core level for the hybrid nanosheet before photoirradiation (black), photooxidized SiO<sub>2</sub> (blue), and thermally oxidized SiO<sub>2</sub> film (red). (b) AR-XPS spectra on the takeoff angle from  $15^{\circ}$  to  $75^{\circ}$  of a photooxidized SiO<sub>2</sub> film.

identical to that of a Si-O-Si network structure of conventional SiO<sub>2</sub> film.<sup>10</sup> The experimental results provide an effective mechanism for photochemical reaction induced by deep UV irradiation. The energy of the deep UV lamp is sufficient for any bond energy in p(DDA/SQ); it results in photocleavage in the hybrid nanosheet. After cleavage of the bond, extremely reactive Si, O<sub>2</sub>, and O species will immediately and mutually react to form silicon dioxide and a small amount of silanol.<sup>11</sup> It is noteworthy that no change in peak position (1065  $\text{cm}^{-1}$ ) was observed with varying the number of deposited layers.

X-ray photoelectron spectroscopy (XPS)<sup>12</sup> confirmed that the p(DDA/SQ) nanosheet is a good precursor for high-quality SiO<sub>2</sub> film formation at room temperature (Figure 2). The XPS spectra of the p(DDA/SQ) nanosheet, photooxidized SiO<sub>2</sub> film, and thermally oxidized SiO<sub>2</sub> film (Furuuchi Chemical Corp.) displayed strong Si 2p peaks at 102.48 eV, 103.25 eV, and 103.23 eV, respectively. The Si 2p peak (102.48 eV) of the hybrid nanosheet before photooxidation is based on Si groups of POSS. That peak was shifted to the Si 2p core levels of the typical SiO<sub>2</sub> film by photoirradiation. The Si 2p fwhm of photooxidized SiO<sub>2</sub> film showed 1.96 eV, in excellent agreement with that of the thermally oxidized SiO<sub>2</sub> film, which was purchased as a reference. Furthermore, to investigate the depth profile of the oxidation states, the angle-resolved XPS spectra were measured by changing the takeoff angle from 15° to 75°. No change in either of the Si 2p peak positions or fwhm of the photooxidized SiO<sub>2</sub> film was observed through the whole range, although the increase of the Ca 2p peak based on CaF2 substrate as the takeoff angle increased. These results from the surface, bulk, and interface of the films suggest that the Si<sup>4+</sup> concentration is uniform at all depths within the film, and that the photooxidized SiO<sub>2</sub> film is of high quality with no suboxide species. For typical SiO<sub>2</sub> film formation from Si by photo, ozone, or thermal oxidation, the transition layer formation is an unavoidable result at the SiO<sub>2</sub>/Si interface.<sup>1,13</sup> This work provides a straightforward procedure for creating a high-quality ultrathin SiO<sub>2</sub> film. The p(DDA/SQ) nanosheets act as a versatile template for the SiO<sub>2</sub> film. Using them, it is possible to coat any substrate completely with p(DDA/SQ) nanosheets. The photooxidized SiO<sub>2</sub> film has another outstanding feature: photopatterning. Figure 3 portrays AFM images of photopatterned SiO<sub>2</sub> films. The SiO<sub>2</sub> film emerges in negative tone mode. The line width is identical to that of the

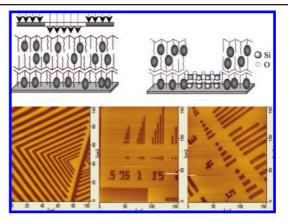


Figure 3. AFM images of the photopatterened SiO<sub>2</sub> films.

photomask. The direct writing of SiO<sub>2</sub> patterns with high resolution offers a wide range of possibilities for practical applications.

In summary, we have demonstrated a photoinduced high-quality SiO<sub>2</sub> film produced from a p(DDA/SQ) hybrid nanosheet. The method is simple and available at room temperature; photooxidation process requires no initiator or annealing process involving postbaking. We obtained an ultrathin, 1.6-nm-thick high-quality SiO<sub>2</sub> film, as demonstrated above. The film thickness is predictable and easily reproduced. Because it depends strongly on the amount of SQ in the hybrid nanosheet, the SiO<sub>2</sub> thickness can be determined easily by controlling the precursor hybrid nanosheet. This approach is a practical means to develop flexible optical and electrical devices.

Acknowledgment. This work was supported by Grants-in-Aid for Scientific Research (S) (no. 17105006) from the Japan Society for the Promotion of Science. This work was also supported by the Special Education and Research Expenses from the Ministry of Education, Culture, Sports, Science and Technology.

Supporting Information Available: Experimental details. This material is available free of charge via the Internet at http://pubs.acs.org.

#### References

- Green, M. L.; Gusev, E. P.; Degraeve, R.; Garfunkel, E. L. J. Appl. Phys. 2001, 90, 2057. (1)
- Sirringhaus, H.; Kawase, T.; Friend, R. H.; Shimoda, T.; Inbasekaran, M.; (2)Wu, W.; Woo, E. P. Science 2000, 290, 2123. Veres, J.; Ogier, S.; Lloyd, G.; de Leeuw, D. Chem. Mater. 2004, 16, 4543. Yoon, M. H.; Yan, H.; Facchetti, A.; Marks, T. J. J. Am. Chem. Soc. 2005, 127, 10388.
   Bergonzo, P.; Boyd, I. W. J. Appl. Phys. 1994, 76, 4372. Nagasawa, H.;
- Kitajima, H.; Kitayama, D.; Okamoto, Y.; Ikoma, H. Jpn. J. Appl. Phys., Part 2 1995, 34, L1103. Nishiguchi, T.; Nonaka, H.; Ichimura, S.; Morikawa, Y.; Kekura, M.; Miyamoto, M. Appl. Phys. Lett. 2002, 81, 2190.
- Godet, C.; Etemadi, R.; Clerc, C. *Appl. Phys. Lett.* **1996**, *69*, 3845.
   (4) Liu, P.; Wu, Y. L.; Li, Y. N.; Ong, B. S.; Zhu, S. P. *J. Am. Chem. Soc.* **2006**, *128*, 4554. Bao, Z. N.; Kuck, V.; Rogers, J. A.; Paczkowski, M. A. Adv. Funct. Mater. 2002, 12, 526
- Brinkmann, M.; Chan, V. Z. H.; Thomas, E. L.; Lee, V. Y.; Miller, R. D.; Hadjichristidis, N.; Avgeropoulos, A. Chem. Mater. 2001, 13, 967. Ouyang, M.; Yuan, C.; Muisener, R. J.; Boulares, A.; Koberstein, J. T. Chem. Mater. 2000, 12, 1591.
- (6) Mirley, C. L.; Koberstein, J. T. Langmuir 1995, 11, 1049.
  (7) Mitsuishi, M.; Matsui, J.; Miyashita, T. Polym. J. 2006, 38, 877. Ishifuji,
- M.; Mitsuishi, M.; Miyashita, T. Appl. Phys. Lett. 2006, 89, 011903. Mitsuishi, M.; Zhao, F.; Kim, Y.; Watanabe, A.; Miyashita, T. Chem. Mater. 2008. 20. 4310.
- (9) Liou, H. C.; Pretzer, J. *Thin Solid Films* **1998**, *335*, 186.
   (10) Boyd, I. W.; Wilson, J. I. B. *J. Appl. Phys.* **1982**, *53*, 4166.
- (11) Boyd, I. W.; Craciun, V.; Kazor, A. Jpn. J. Appl. Phys., Part 1 1993, 32, 6141. Chan, V. Z. H.; Hoffman, J.; Lee, V. Y.; Iatrou, H.; Avgeropoulos, A.; Hadjichristidis, N.; Miller, R. D.; Thomas, E. L. Science 1999, 286, 1716.
- (12) Himpsel, F. J.; McFeely, F. R.; Talebibrahimi, A.; Yarmoff, J. A.; Hollinger, G. Phys. Rev. B 1988, 38, 6084. Holl, M. M. B.; Lee, S. H.; McFeely, F. R. Appl. Phys. Lett. 1994, 65, 1097.
- (13) Iwata, S.; Ishizaka, A. J. Appl. Phys. 1996, 79, 6653.
- JA803852W